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Directed polymer in random media in two dimensions: Numerical study of the aging dynamics

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Following a recent work by Yoshino $[J.$ Phys. A 29 , 1421 (1996) , we study the aging dynamics of a directed polymer in random media, in $1+1$ dimensions. Through temperature quench and temperature cycling numerical experiments similar to the experiments on real spin glasses, we show that the observed behavior is comparable to that of a well-known mean-field spin glass model. The observation of various quantities (correlation function, "clonation" overlap function, etc.) leads to an analysis of the phase space landscape. $[S1063-651X(97)00704-6]$

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I. INTRODUCTION

The study of directed polymers in random media has triggered a lot of interest and of works (for a recent review, see $[1]$, since it is related to many fields, from the fluctuations of interfaces $\lceil 2 \rceil$ to quantum-mechanical problems in a timedependent random potential $[3]$, or the very topical problem of vortex lines in high-temperature superconductors $[4]$. It also has connections with spin glasses, as has been shown by Derrida and Spohn $[5]$, who studied a mean-field version (on a Cayley tree) of the random polymer, showing the existence of a low-temperature phase similar to the random energy model of Derrida $[6]$.

In finite dimensions $(d+1, \text{ with } d \text{ transverse dimension})$ sions), the existence of a phase transition has been shown for $d \ge 2$ [7–9], whereas the system is always in a lowtemperature phase for $d=1$, and has been called a "baby" spin glass" in $[10]$.

The dynamics of such a model has not been much studied so far. A recent numerical work by Yoshino $[11]$ has made clear the existence of aging for the directed polymer in random media in dimensions $1+1$, with violation of time translation invariance and of the fluctuation-dissipation theorem. The observed behavior for the correlation function is similar to the scaling properties for the simulations of a threedimensional spin glass model $\lfloor 12 \rfloor$. It is analyzed along the lines of a scenario similar to the droplets model for spin glasses $[13,14]$: the polymer moves in a network of "tubes" where its probability of presence (calculated at equilibrium with a transfer matrix method) is high. The network has a quite complicated spatial structure, and the tubes form loops of various sizes. The dynamics consists then of rapid fluctuations inside the tubes (acting as traps) combined with thermally activated jumps between different tubes. These thermal excitations are compared to the droplets excitations. Besides, the fact that the loops display a broad distribution of sizes $[15]$ induces a broad distribution of relaxation times, and thus aging.

On the other hand, Cugliandolo, Kurchan, and Le Doussal, following the study of the aging dynamics of a meanfield spin glass model $[16]$, and of a particle in an infinite dimensional random potential $[17,18]$, have proposed an analytical treatment of the long times off equilibrium dynamics of an elastic manifold embedded in an infinite dimensional space, in the presence of a quenched random potential $[19]$ (the statics of such a model has been studied by Mézard and Parisi with a replica variational Gaussian approximation, becoming exact in this infinite dimensional limit $|20|$. This corresponds here to an infinite *d*. The finite dimension of the manifold leads to the study of the relaxation of its Fourier modes *k*. The two-times correlation and response functions $C_k(t,t')$ and $r_k(t,t')$ satisfy dynamical equations where two regimes can be separated, as for the case of the particle $[17,18]$: a stationary regime, and a regime displaying aging, where the properties of equilibrium dynamics (namely, time

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translation invariance and fluctuation dissipation theorem) are violated. Various equations can be written for the longtimes behavior of the functions, for example, the measure of the violation of the fluctuation dissipation theorem $X(C)$ does not depend on *k*.

We will here focus on the same model as $\lfloor 11 \rfloor$; after describing this model, the used dynamics, we perform several numerical experiments for aging dynamics. Preparing the polymer in its ground state, we also show that it can display stationary dynamics, and compare the two kinds of dynamics in order to analyze the phase space landscape.

A. Model

The polymer is defined on a square lattice of linear size *N*: it consists of *N* monomers lying on the sites $\{(i, x_i), i = 1, \ldots, N\}$, and the Hamiltonian is

$$
H[\{x_i\}] = \sum_{i=2}^{N} [|x_i - x_{i-1}| + V(i, x_i)],
$$
 (1)

where *V* is a Gaussian random potential, with zero mean and variance σ , uncorrelated from site to site. The first term gives the elastic energy; moreover, the steps $|x_i - x_{i-1}|$ are restricted to 0 or 1, and one of the extremities is fixed, so that a Boltzmann measure can be defined: $x_1 = N/2$.

The transfer matrix method can be used to study the statics of this model [10]. In particular, the Edwards-Anderson (EA) parameter can be calculated, following $[10]$: we take two replicas of the polymer, with the same realization of the disordered potential; if $Z(x, y, L)$ is the partition function for the pairs of polymers arriving at transverse coordinates *x* and the pairs of polymers arriving at transverse coordinates x and y after *L* steps (or monomers), and if $\tilde{Z}(x,y,q,L)$ is the partition function restricted to such pairs having an overlap *q*, it is possible to write recursion relations in *L* for *Z* and $Y(x, y, L) = \sum_{q} q \overline{Z}(x, y, q, L)$, and therefore to evaluate

$$
\lim_{L \to \infty} \frac{1}{L} \frac{\sum Y(x, y, L)}{\sum Z(x, y, L)},
$$
\n(2)

and to average it over disorder to obtain q_{EA} . We will use this value to check some expected long time limits of dynamical quantities (see below).

The directed polymer is evolving with Monte Carlo dynamics in a heat bath in the following way: a monomer and a move are chosen at random, and the move is performed with probability min $[1, \exp(-\beta \Delta E)]$ (Metropolis algorithm), β being the inverse temperature, and ΔE the change of energy involved. One Monte Carlo step consists in *N* such tries.

B. Numerical experiments

The transfer matrix method allows us to find the ground state in a given realization of a potential; we have studied the dynamics in two cases: the initial configuration is either taken at random, or as the ground state. The polymer, therefore, coming from an infinite temperature or from a zero temperature thermalized state, is then free to evolve at the temperature of the heat bath. We then measure the evolution of the energy, the two-times correlation function defined by

$$
C(t_w + t, t_w) = \left\langle \sum_{i=1}^{N} \delta_{x_i(t_w + t), x_i(t_w)} \right\rangle,
$$
 (3)

where $\langle \rangle$ is a mean over thermal noise and the overline denotes a mean over realizations of the quenched disorder, and the overlap between two copies of the polymer evolving in the same realization of the potential, with the same thermal noise until t_w , and then decoupled $[21-24]$ (a process called "clonation" in $[21]$); if these copies are labeled by (1) and (2) , this overlap is

$$
Q_{t_w}(t_w + t, t_w + t) = \left\langle \sum_{i=1}^{N} \delta_{x_i^{(1)}(t_w + t), x_i^{(2)}(t_w + t)} \right\rangle.
$$
 (4)

Besides, we have performed temperature cycling experiments in the same way as for real spin glasses $[25]$. Most of the runs have been made with a polymer of length $N = 500$, and some with $N = 800$: no finite size effects were seen for the used simulation times.

II. AGING DYNAMICS

A. Quench at initial time

The initial configuration of the polymer is chosen at random, with the constraint $|x_i - x_{i-1}| = 0$ or 1; then it evolves at a fixed temperature. It has already been observed $[11]$ that the correlation function $C(t_w + t, t_w)$ previously defined displays aging behavior: it depends explicitly on t_w and t (see Fig. 1); as the system ages, it becomes more rigid, in the sense that it evolves slower and gets away from itself always slower.

For times *t* much lower than t_w , the dynamics has the characteristics of equilibrium dynamics: $C(t_w + t, t_w)$ depends only on *t* (time translation invariance), and it has been checked numerically $\begin{bmatrix} 11 \end{bmatrix}$ that the fluctuation-dissipation theorem is valid. Besides, we have also checked the validity of the relation $Q_{t_w}(t_w + t, t_w + t) = C(t_w + 2t, t_w)$ [24]. In this regime, we have therefore a quasiequilibrium dynamics. Some well-known spin glass models (like, for example, the *p*-spin spherical model $[16]$ or the random manifold $[17,19]$ present a correlation function decaying from 1 to q_{EA} , with a power law approach to the q_{EA} plateau: $q_{EA} + At^{-\nu}$.

We have been able to fit the results of the simulations with such form, using the values of q_{EA} obtained by the transfer matrix method: this fit therefore uses only two parameters, and not three (see Fig. 2 for the values, and Fig. 1 for examples of the fits). The obtained values of ν (typically in the range $0.1-0.2$) are displayed in Fig. 3.

For times *t* comparable to or bigger than t_w , the correlation function $C(t_w + t, t_w)$ depends explicitly on t_w : the dynamics is no more time translation invariant. For $t \geq t_w$, it decays as a power law,

$$
C(t_w+t,t_w) \approx f(t_w,T) \left(t/t_w\right)^{-\lambda}.
$$
 (5)

We are therefore in the presence of a weak-ergodicity breaking behavior, $\lim_{t\to\infty} C(t_w + t, t_w) = 0$ [26]. The obtained values of λ are comparable to the values of ν (see Fig. 3), and both exponents are increasing functions of temperature, like in real spin glasses $[25]$.

FIG. 1. (a) From bottom to top, $C(t_w + t, t_w)$ versus *t* for $t_w = 50$, 500, 5000, and 100 000, at $T = 1$, and the fit $C_{as}(t) = 0.65 + 0.25t^{-0.08}$. (b) $C(t_w + t, t_w)$ versus *t* for $T = 3$ and t_w = 2000, 50000, and 100 000, and $C_{as}(t)$ = 0.24 + 0.545*t*^{-0.185}. In both cases, $q_{EA}(T=1) = 0.65$ and $q_{EA}(T=3) = 0.24$ have been obtained by the transfer matrix method.

It is important to note that the ν exponent is different from the *x* exponent studied by Yoshino [11]: this *x* is obtained by the scaling form

 $C(t_w + t, t_w) \approx t^{-x}$ (6)

FIG. 3. Exponents λ (for *T*=1,2,3,4,5) and *v* (for *T*=1,2,3) versus temperature.

for the $t \ll t_w$ part, with a global form

$$
C(t_w + t, t_w) = t^{-x} \Phi(t/t_w).
$$
 (7)

If we take the limit $\lim_{t\to\infty} \lim_{t\to\infty}$ we obtain the same behavior, but the opposite order of limits, $\lim_{t\to\infty}\lim_{t\to\infty}C(t_w+t,t_w)$ yields 0, in contradiction with the expected static limit

$$
\lim_{t \to \infty} \lim_{t_w \to \infty} C(t_w + t, t_w) = q_{\text{EA}},
$$
\n(8)

which is also obtained by the form $q_{EA} + At^{-\nu}$.

With numerical data, however, it is difficult to prefer one of these forms, and much longer simulations (much bigger values of t_w) would be necessary.

If we now look at the overlap between two replicas separated at t_w , Q_{t_w} , it seems that this function has a finite limit at large times t (see Fig. 4), with a value compatible with the value of q_{EA} , for big enough t_w . This constatation puts this model in the class I of the classification of $[24]$, which includes domain-growth models, and the $p=2$ spherical p -spin model [23]: it indicates that the evolution in phase space takes place in "corridors" [21], of size q_{EA} . On the contrary, the case of the manifold embedded in an infinite dimensional space yields $\lim_{t_w \to \infty} \lim_{t \to \infty} Q_{t_w}(t_w + t, t_w)$ $(t + t) = 0$, and therefore belongs to type II, which probably indicates a much more complex phase-space landscape, and occurs also, for example, for the *p*-spin spherical model with $p \geq 3$ [24].

B. Temperature cycling experiments

A spin glass quenched under its transition temperature, and then submitted to temperature changes, shows a very puzzling behavior (see $[25]$ for a review, and references therein) that we briefly describe now. After a quench at time $t=0$, the temperature cycle is as follows (see Fig. 5): the temperature is *T* from $t=0$ to $t=t_1$, then $T+\Delta T$ from $t = t_1$ to $t = t_2$, and again *T* after $t = t_2$; ΔT can be negative as well as positive. For a positive ΔT , one observes a reinitialization of the dynamics, with, e.g., for a thermoremanent magnetization, a relaxation after t_2 identical to the one obtained after a quench at $t=0$ and a waiting time $t_{\text{eff}} = t_w - t_2$. On the contrary, for a negative ΔT , the system

FIG. 4. (a) $Q_{t_w}(t_w + t, t_w + t)$ for $t_w = 50, 500, 5000$, at $T=1$, in logarithmic scale and (b) $Q_{t_w}(t_w+t,t_w+t)$ versus *C*($t_w + t$, t_w) for $t_w = 50$, at *T*=1. At *T*=1, $q_{EA} \approx 0.65$.

keeps memory of its evolution, and its relaxation corresponds to an effective age between t_w and $t_w - t_2 + t_1$.

We have performed these jumps numerically with *T*=2, ΔT =1,2,-1, t_1 =500, and t_2 =1500. We have then monitored the evolution of the energy of the polymer, as well as $C_{\text{jump}}(t_w + t, t_w)$ with $t_w = 2000$, comparing these quantities with the ones obtained for $\Delta T=0$ (constant temperature).

It is clear (Fig. 6) that $C_{\text{jump}}(t_w + t, t_w)$ corresponds to a certain $C(t_{\text{eff}}+t,t_{\text{eff}})$, t_{eff} being an effective age for the system, depending on $t_2 - t_1$ and on ΔT ; t_{eff} is less than t_w for a negative ΔT , which means that the dynamic has been slowed down by the time spent at a lower temperature; here we estimate $t_{\text{eff}} = 1000 = t_w - t_2 + t_1$ [for $\Delta T = -1$, the data
corresponding to $C_{\text{imm}}(2000 + t, 2000)$ with to $C_{\text{jump}}(2000 + t, 2000)$ with $t_1 = 500$, $t_2 = 1500$ are superimposed on the curve $C(1000+t,1000)$. For smaller values of ΔT , or longer times t_2-t_1 , t_{eff} can be bigger than $t_w-t_2+t_1$: the time spent at $T + \Delta T$ can contribute a little to the aging. For positive values of ΔT , t_{eff} is bigger than t_w (in Fig. 6, $t_{\text{eff}} = 3000$), showing that the time spent at $T + \Delta T$ has contributed to the approach to equilibrium more than the same time at *T*.

The behavior of the directed polymer is therefore symmetrical for positive or negative variations of temperature. No reinitialization of dynamics is found. This type of behavior is similar to the one observed for mean-field spin glasses

FIG. 5. Temperature cycles.

 $[25,23]$, and thus very different from the one observed in real experiments on spin glasses.

It should be remarked that real experiments deal with a response function, whereas we are monitoring correlation functions. However, numerical simulations of a three-

FIG. 6. Results of the temperature cycling experiments: The symbols show $C(t_w + t, t_w)$ at constant temperature $T=2$ for t_w =1000 (squares), t_w =2000 (crosses), and t_w =3000 (diamonds). The temperature cycles are done and with $t_1 = 500$, $t_2 = 1500$, and t_w =2000, at *T*=2, and yield $C_{\text{jump}}(t_w + t, t_w)$: the corresponding curves are superimposed on $C(1000+t,1000)$ for $\Delta T = -1$ and on *C*(3000+*t*,3000) for $\Delta T = 1$.

FIG. 7. $C(t_w + t, t_w)$ versus *t*, in logarithmic scales: comparison of the evolutions for various initial conditions. For each figure, the three lower curves correspond to random initial conditions [(a) for $T=1$, $t_w=500$, 5000, 100 000, and (b) for $T=2$, t_w $=10000$, 50000, 100000], while three curves corresponding to the system being prepared in its ground state at initial time are superimposed onto each other (with the same waiting times t_w as for the lower curves), showing that $C(t_w + t, t_w)$ depends only on *t* in this case.

dimensional Ising spin glass model have shown also for the correlation functions $[27]$ a partial reinitialization of the dynamics for positive ΔT , and asymmetric outcomes of numerical experiments with positive or negative temperature cycles. Nothing of this kind is found here.

III. RELAXATION FROM THE GROUND STATE

The transfer matrix method allows us to find the ground state of the polymer, given a realization of the potential. We then let the polymer evolve at temperature *T* as before.

The measure of the correlation function $C(t_w + t, t_w)$ for various t_w and t shows in this case a simple t dependence: the system is time translation invariant (Fig. 7). Besides, $C(t_w + t, t_w)$ does not seem to go to zero for large *t*, or at least stays well above the correlation at the same times, for a system with random initial condition, for the accessible times: no power law decay $t^{-\lambda}$ is found. It seems that ergodicity is really, and not simply weakly, broken.

We have also studied the dynamics for an initial configuration of energy close to the ground state energy, but spa-

FIG. 8. (a) Evolution of the energies as a function of time, for $T=2$, for random initial conditions (upper curve), or for a system prepared in its ground state (lower curve). (b) $E_{al} - E_f$ versus time, for $T=2$, in logarithmic scale (symbols); the straight line corresponds to the power law $t^{-0.15}$.

tially well separated. In this case, we observe a similar behavior, with a time translation invariant correlation function.

Such behavior has been observed, for example, in the TAP states of the *p*-spin spherical model with $p \ge 3$ [28].

IV. ENERGY

We have monitored the evolution of the energy density E_{al} of the polymer; for a random initial condition the initial energy is high, so the initial behavior is a fast decay, followed by a much slower evolution.

When the polymer is prepared in its ground state, on the contrary, its energy E_f grows quickly because of the thermal bath, and then stays constant.

We show in Fig. 8 the difference between the energy densities for both situations, in logarithmic scales: this plot shows that the evolution is compatible with a power law decay of E_{al} towards E_f . All the aging dynamics, therefore, take place at higher energy densities than those of the low lying states, where the dynamics is stationary.

V. DISCUSSION

Whereas the previous analysis of the dynamics made by Yoshino was based on an analogy with the droplets model $[13,14]$, we focus here more on a phase space analysis.

It must, however, be clear that, as for many numerical simulations, especially for glassy systems, the available time scales remain quite small, and that the results should therefore be considered as tendencies, indications of behavior. They allow us, with these precautions, to present the following analysis.

The observed behavior is quite similar to the one found for the spherical *p*-spin model with $p=2$: the dynamic consists in a slow search of the ground state, with a slowly decaying energy. There exists many states with low energy, but the polymer is not able to find them, during its aging dynamics: it remains at higher energy density; on the contrary, if it is put in one of these configurations, it stays trapped and has a stationary dynamics.

The behavior of the overlap of two copies of the system, having the same configurations until a certain time, and then decoupled, shows besides that the evolution takes place in some kind of ''gutters'' in phase space. The results of temperature cycling experiments also show that the directed polymer in $1+1$ dimensions is a much simpler system than real spin glasses. These two results are probably related, and also in agreement, concerning the relative simplicity of the phase space, with the fact $[11]$ that the response to a tilt field applied at the end of the polymer does not display any aging; however, it would be nice to measure the response to a spatially sinusoidal field, and the relaxation after cutting the field, to check if such a relaxation has aging, and how it depends from the wavelength of the field. However, the sample-to-sample fluctuations of the response functions are very important, and such measurements are therefore very difficult.

These behaviors are in fact intermediate between the $p=2$ spherical *p*-spin model (analogous to domain growth [23]) and the $p \ge 3$ case, which, despite being a mean-field model, displays a much more complicated behavior with long term memory and a very complex phase space $[16]$.

It would certainly be very interesting to study the directed polymer dynamics in higher dimensions: these new dimensions could provide a way to avoid energy barriers by going around them. The appearance of these entropy barriers $[29-$ 32] could yield new interesting effects and a richer dynamic in a more complex phase space.

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